# ON GLASS TRANSITION TEMPERATURE (T<sub>G</sub>) FOR CONFINED THIN FILMS

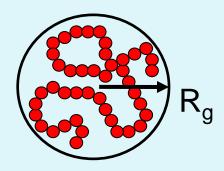
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#### **BACKGROUND - CONFINEMENT**

#### **BULK**

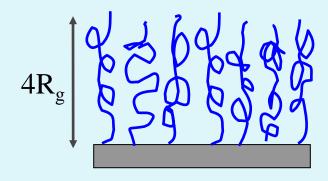


Polymer chains in bulk have random coil conformation with average size of radius of gyration ( $R_g$ ).

#### **POLYMER UNDER CONFINEMENT**



Polymer thin film

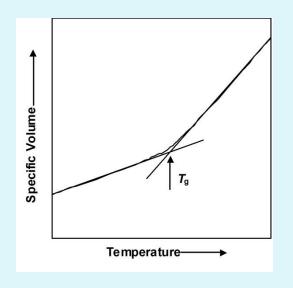


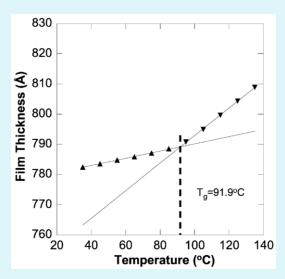
Polymer brush

➤ High grafting density (~0.5 chains/nm²) leads to chains stretching normal to the surface, creating a highly confined system.

### WHAT IS T<sub>G</sub>?

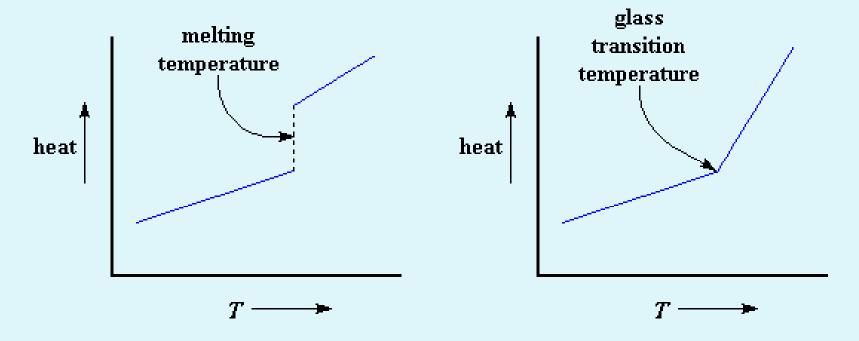
• Glass transition temperature  $(T_g)$  – the point of transition at which amorphous materials change from brittle, glass state to soft, rubber state.





- Physical properties (specific volume, heat capacity, viscosity, thermal expansion coefficient) change.
- Thermal expansion coefficient change is directly proportional to the thickness change for the thin films.

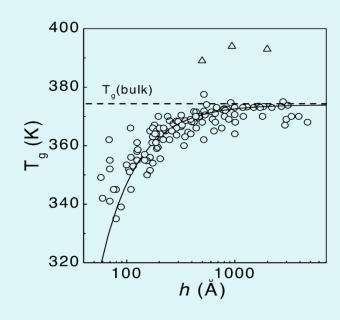
# T<sub>G</sub> VS. MELTING POINT



A heat vs. temperature plot for an crystalline polymer, on the left; and a amorphous polymer on the right.

#### **MOTIVATION**

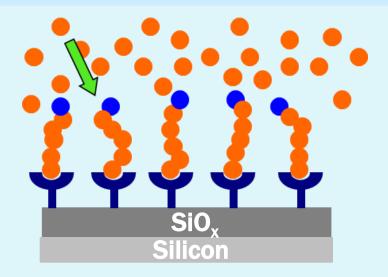
- For Polystyrene (PS) thin films the T<sub>g</sub> decreases as films get thinner.
- Proven recently that tethering alters the surface dynamics of the chains due to extreme lateral and vertical confinement.<sup>1</sup>
- How does dense tethering alter the T<sub>g</sub> of the thin films?
- Fundamental Science
- Industry



J.A. Forrest et al., The Glass Transition in Thin Polymer Films, *Advances in Colloid and Interface Science*, 2001, 94, 167.

#### **EXPERIMENTAL SETUP**

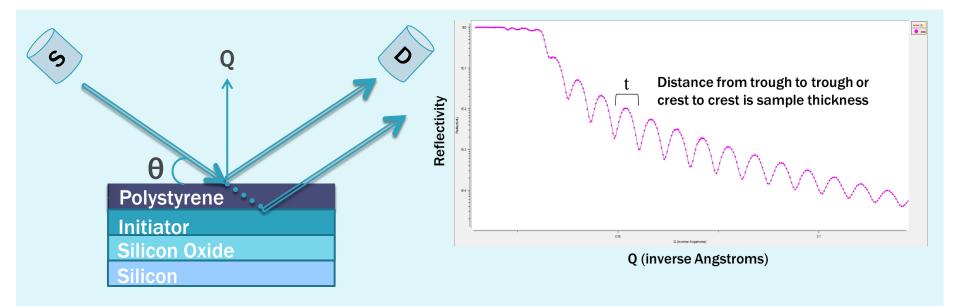
- Reflectivity scans on high-density PS brushes
- Annealed for two hours at 120°C



- Bromine
- Polystyrene
- 11-(2-bromo-2-ethyl)propionyloxy undecyltrichlorosilane (initiator layer)

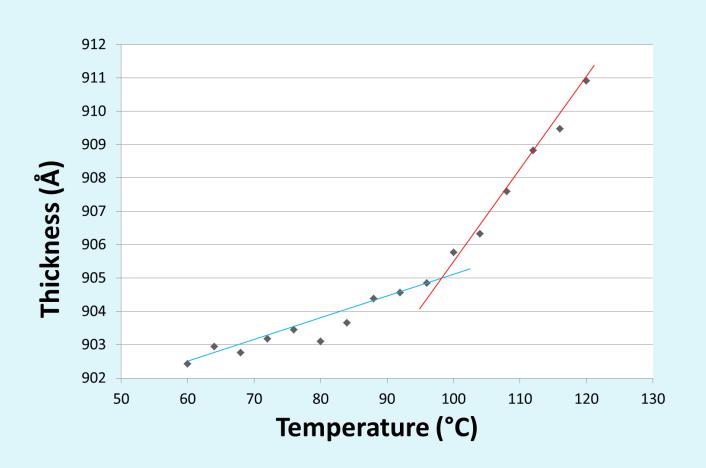


#### X-RAY REFLECTIVITY



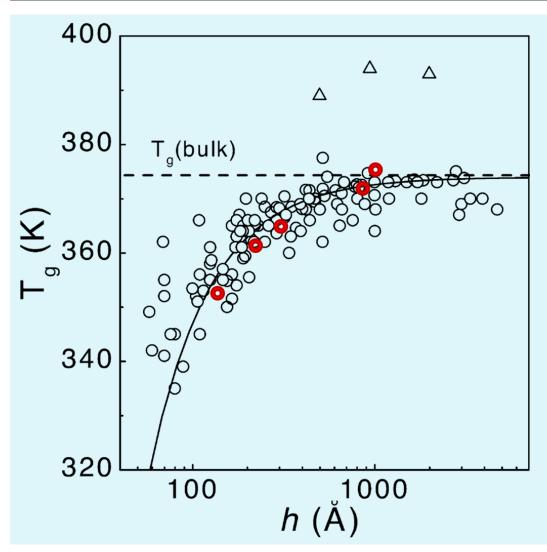
- Interference of these partially reflected x-ray beams creates a reflectometry pattern
- Kiessig fringes (maxima) occur because interference of waves can be constructive or destructive,  $t=2\pi/\Delta q$

## T<sub>G</sub>, GLASS TRANSITION TEMPERATURE



| Temperature | Thickness          |
|-------------|--------------------|
| 25          | 906.17             |
| 40          | 903.581            |
| 60          | 902.427            |
| 64          | 902.941            |
| 68          | 902.76             |
| 72          | 903.169            |
| 76          | 903.447            |
| 80          | 903.089            |
| 84          | 903.652            |
| 88          | 904.378            |
| 92          | 904.558            |
| 96          | 904.846            |
|             |                    |
| 100         | 905.763            |
| 104         | 906.321            |
| 108<br>112  | 907.587<br>908.825 |
| 112         | 909.466            |
| 120         | 910.913            |
|             | 320.020            |

# T<sub>G</sub> DECREASES AS FILM THICKNESS DECREASES



- 14 nm (140 Å) 351 K
- 22 nm (220 Å) 359 K
- 31 nm (310 Å) 363 K
- 90 nm (900 Å) 370 K
- 100 nm (1000 Å) 373 K

J.A. Forrest et al., The Glass Transition in Thin Polymer Films, *Advances in Colloid and Interface Science*, 2001, 94, 167.

#### **SUMMARY**

- There is indeed a trend for thinner films to have lower T<sub>g</sub> values.
- The reduction in T<sub>g</sub> for thinner films implies that these films can be processed at lower temperatures and large amount of energy can be saved.
- We need thinner (< 10 nm) PS brush samples to verify if tethering alters the T<sub>g</sub>. So far our data suggests there is no difference between tethered and untethered chains.

#### **ACKNOWLEDGMENTS**

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